

equation.<sup>34,35</sup> Due to these reasons we regard  $\xi$  as an adjustable parameter. As shown by the chain line in Figure 7, the best-fit curve has been obtained by taking  $\xi = 0.482$ .

From the value of  $\Delta$  for the branched PDCPO, the distribution of the chain lengths of branches, including the main chain, can be assessed in terms of eq 16. In this calculation,  $\xi$  and  $\alpha$  are taken to be 0.482 and 0.62, respectively. The values of  $M_w(b)/M_n(b)$  thus calculated are listed in Table I. As expected,  $M_w(b)/M_n(b)$  is much larger than  $M_w/M_n$  for the whole molecule and increases with molecular weight.

**Acknowledgment.** We thank Professor W. H. Stockmayer for instructive suggestions to revise the manuscript. We also thank Professor A. M. North for helpful discussions.

**Registry No.** Poly[oxy(2,6-dichloro-*p*-phenylene)], 26023-26-7; 2,6-dichlorophenol homopolymer, 25511-62-0.

## References and Notes

- (1) Stockmayer, W. H. *Chem. Soc. Rev.* **1966**, *15*, 539.
- (2) North, A. M. *Chem. Soc. Rev.* **1972**, *1*, 49.
- (3) Zimm, B. H. *J. Chem. Phys.* **1956**, *24*, 269.
- (4) Stockmayer, W. H.; Baur, M. E. *J. Am. Chem. Soc.* **1964**, *86*, 3485.
- (5) Rouse, P. E. *J. Chem. Phys.* **1953**, *21*, 1272.
- (6) Bueche, F. *J. Chem. Phys.* **1954**, *22*, 603.
- (7) de Gennes, P.-G. *J. Chem. Phys.* **1971**, *55*, 572.
- (8) Doi, M.; Edwards, S. F. *J. Chem. Soc., Faraday Trans. 2* **1978**, *74*, 1789, 1802, 1818.
- (9) Baur, M. E.; Stockmayer, W. H. *J. Chem. Phys.* **1965**, *43*, 4319.
- (10) Stockmayer, W. H.; Burke, J. J. *Macromolecules* **1969**, *2*, 647.
- (11) Jones, A. A.; Stockmayer, W. H.; Molinari, R. J. *J. Polym. Sci., Polym. Symp.* **1976**, No. 54, 227.
- (12) North, A. M.; Phillips, P. J. *Trans. Faraday Soc.* **1968**, *64*, 3235.
- (13) Hirose, M.; Yamakawa, N.; Araki, K.; Imamura, Y. *Rep. Prog. Polym. Phys. Jpn.* **1977**, *20*, 117.
- (14) Blanchard, H. S.; Finkbeiner, H.; Russell, G. A. *J. Polym. Sci.* **1962**, *58*, 469.
- (15) Stamatoff, G. S. French Patent 1 403 987, May 17, 1965.
- (16) Carr, B. G.; Harrod, J. F.; van Gheluwe, P. *Macromolecules* **1973**, *6*, 498.
- (17) Yamakawa, H. "Modern Theory of Polymer Solutions"; Harper and Row: New York, 1971; Chapter 6.
- (18) Zimm, B. H.; Roe, G. R.; Epstein, L. F. *J. Chem. Phys.* **1956**, *24*, 276.
- (19) Hay, A. S. *Fortsch. Hochpolym. Forsch.* **1967**, *4*, 496.
- (20) Hirose, M.; Ikeda, K.; Imamura, Y. *Nippon Kagaku Kaishi* **1976**, *10*, 1608.
- (21) Staffin, G. D.; Price, C. C. *J. Am. Chem. Soc.* **1960**, *82*, 3632.
- (22) Ouano, A. C. *J. Polym. Sci., Polym. Chem. Ed.* **1974**, *12*, 1151.
- (23) Kotaka, T. *J. Appl. Polym. Sci.* **1977**, *21*, 501.
- (24) Harrod, J. F.; van Gheluwe, P.; Kisakurek, D.; Shaver, A. *Macromolecules* **1981**, *14*, 565.
- (25) Adachi, K.; Ohta, K.; Kotaka, T. *Polymer*, submitted.
- (26) McCrum, N. G.; Read, B. E.; Williams, G. "Anelastic and Dielectric Effects in Polymeric Solids"; Wiley: New York, 1967.
- (27) Mashimo, S.; Chiba, A. *Polym. J.* **1971**, *55*, 572.
- (28) Bates, T. W.; Ivin, K. J.; Williams, G. *Trans. Faraday Soc.* **1967**, *63*, 1964.
- (29) Stockmayer, W. H.; Matsuo, K. *Macromolecules* **1972**, *5*, 766.
- (30) Stark, J. G.; Wallace, H. G. "Chemistry Data Book, SI Edition"; John Murray: London, 1970.
- (31) Minkin, V. I.; Osipov, O. A.; Zhdanov, Yu. A. "Dipole Moments in Organic Chemistry"; Plenum Press: New York, 1970.
- (32) Zimm, B. H.; Stockmayer, W. H. *J. Phys. Chem.* **1949**, *53*, 1301.
- (33) Zimm, B. H.; Kilb, R. W. *J. Polym. Sci.* **1959**, *37*, 19.
- (34) Dissado, L. A.; Hill, R. M. *Nature (London)* **1979**, *279*, 695.
- (35) Williams, G.; Cook, M.; Hains, P. J. *J. Chem. Soc. Faraday Trans. 2* **1972**, *68*, 1045.

## Notes

### Concentration Dependence of the Translational Diffusion Coefficient of Unperturbed Flexible Chains

YOSHISUKE TSUNASHIMA\* and NORIO NEMOTO

Institute for Chemical Research, Kyoto University, Uji, Kyoto-fu 611, Japan. Received March 24, 1983

The dynamic light scattering (DLS) technique provides direct and highly accurate measurements of the concentration dependence of the diffusion coefficient of a polymer in solution in comparison with other techniques such as sedimentation diffusion (ultracentrifuge, SD) or conventional diffusion over a wide range of polymer molecular weight  $M$ . With the results, the theories on the coefficient  $k_D$  for flexible chains in dilute solutions can be discussed in detail for the polymer models used and the treatment of the hydrodynamic interactions. In this note, we sum up the experimental results of  $k_D$  for linear, narrow-molecular-weight polystyrenes (PS) at the  $\theta$  temperature, including our recent data,<sup>1</sup> and point out a weaker molecular weight dependence of  $k_D^\theta$  than expected from the theoretical expressions ( $k_D^\theta \propto M^{1/2}$ ) of Pyun and Fixman,<sup>2</sup> Yamakawa<sup>3</sup> and Imai,<sup>4</sup> or the recent report of Mulderije.<sup>5</sup>

We may expand the diffusion coefficient  $D$  for monodisperse linear polymers as a function of concentration  $c$  ( $\text{g cm}^{-3}$ )

$$D = D_0(1 + k_D^\theta c + \dots) \quad (1)$$

where  $D_0$  is the translational diffusion coefficient at infinite dilution and proportional to  $M^{1/2}$ . The theoretical predictions available at present for  $k_D^\theta$  may be expressed as

$$k_D^\theta = -(k_f^\theta + \bar{v}) \quad (2)$$

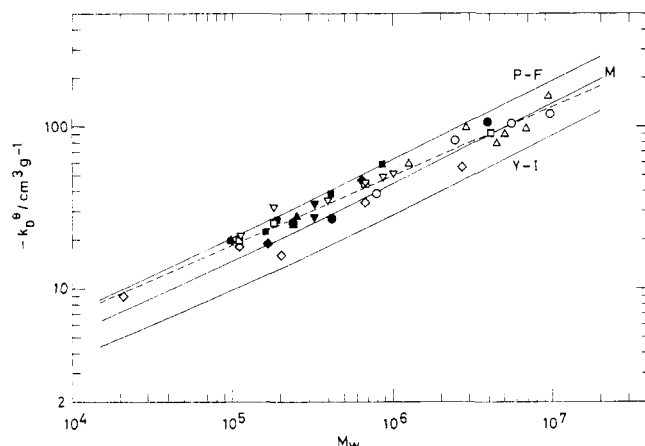
with

$$k_f^\theta = BN_A v_H / M = B'M^{1/2} \quad (3)$$

where  $\bar{v}$  is the partial specific volume of the polymer,  $k_f^\theta$  the coefficient of the concentration dependence of the friction coefficient,  $N_A$  the Avogadro number, and  $v_H$  the hydrodynamic volume of the polymer at infinite dilution. The Einstein-Stokes law expression for  $D_0$

$$D_0 = k_B T / 6\pi\eta_0 R_H \quad (4)$$

and the relation at infinite dilution  $v_H = (4\pi/3)R_H^3$  lead to the second equality in eq 3. Here,  $\eta_0$  is the solvent viscosity and  $R_H$  the hydrodynamic radius of the polymer. Thus all the theories available require an  $M^{1/2}$  dependence of  $k_D^\theta$  in the larger  $M$  region because  $\bar{v} \approx 1$  ( $\text{cm}^3 \text{g}^{-1}$ ). The difference in these theories is revealed only in the magnitude of the constant value  $B$  in eq 3. The Yamakawa<sup>3a</sup>-Imai<sup>4</sup> (Y-I) theories for bead-spring model chains lead to  $B = 1$ , while the Pyun-Fixman (P-F) soft-equivalent-sphere model gives  $B = 2.23$ . Mulderije<sup>5</sup> has recently revised the P-F value as  $B = 2.06$ , moreover indicating  $B = 1.60$ , which is based on the study of the hard spherical particles.<sup>6</sup>



**Figure 1.** Coefficient  $k_D^\Theta$  for the concentration dependence of the translational diffusion coefficient plotted against  $M_w$  for linear narrow-molecular-weight polystyrene in  $\Theta$  solvents. The open symbols represent the dynamic light scattering data: (○) Tsunashima et al.;<sup>1</sup> (□) Han;<sup>7</sup> (◇) King et al.;<sup>8</sup> (Δ) Jones and Caroline;<sup>9</sup> (▽) Frost and Caroline.<sup>10</sup> The filled symbols represent the sedimentation data: (●) Homma et al.;<sup>11</sup> (◆) Wales and Rehfeld;<sup>12</sup> (▲) Billick;<sup>13</sup> (▼) Kotaka and Donkai;<sup>14</sup> (■) Mulderije.<sup>15</sup> The broken line is the fitting curve of  $-k_D^\Theta = 0.13M_w^{0.43}$ ; curves Y<sup>3a</sup>-I,<sup>4</sup> P-F,<sup>2</sup> and M<sup>5</sup> are the theoretical ones represented by  $-k_D^\Theta = B'M_w^{1/2} + 1.1$  with  $B' \times 10^2 = 2.73, 6.09$ , and  $4.37$ , respectively.

The experimental  $k_D^\Theta$  data, as we see now, differ from the above predictions. In Figure 1, we show the  $M_w$  (weight-average molecular weight) dependence of  $k_D^\Theta$  for narrow-molecular-weight PS under the  $\Theta$  condition. The data with open symbols were obtained from the DLS technique<sup>1,7-10</sup> (eq 1) and the data with filled symbols were obtained from the SD technique<sup>11-16</sup> by using eq 2. Two facts become clear from this figure. First, no systematic difference in data due to the two experimental techniques is observed. This may indicate that both the intramolecular segment distribution of the individual separate molecules and the spatial distribution of the molecules in solution are not affected by the difference in the two diffusion processes. In other words, the short-time diffusion<sup>17</sup> seems negligible. Second, the  $k_D^\Theta$  data can never be represented by any single value of  $B$  as long as the  $M^{1/2}$  dependence in eq 2 and 3 is adopted. The least-squares fitting of all of these 40 data points leads to a completely different  $M_w$  dependence of  $k_D^\Theta$  from theories and is expressed as

$$-k_D^\Theta = 0.13M_w^{0.43 \pm 0.02} \text{ (cm}^3 \text{ g}^{-1}) \quad (5)$$

The broken line in Figure 1 represents this equation. The power of 0.43 is 14% smaller than the theoretical value of 0.50. Here, the solid curves Y-I, P-F, and M in Figure 1 represent the Y<sup>3a</sup>-I<sup>4</sup> theory, the P-F<sup>2</sup> theory, and the revised P-F theory (Mulderije<sup>5</sup>) with  $B = 1.60$ , respectively, in the form of eq 3. These  $B'$  values are  $B' \times 10^2 = 2.73, 6.09$ , and  $4.37$ , respectively, which were estimated from our experimental result<sup>1</sup> for PS in *trans*-decalin at 20.4 °C ( $\Theta$  temperature):  $D_0M_w^{1/2} = (4.60 \pm 0.08) \times 10^{-5} \text{ cm}^2 \text{ s}^{-1}$ .

As long as the equilibrium or averaged polymer conformation is independent of hydrodynamic interactions, the power law  $k_D^\Theta \propto M^{1/2}$  is predicted irrespective of the

Gaussian (Y) or uniform (P-F) segment distributions of molecules. Recently, Akcasu's theory<sup>17</sup> based on the generalized Langevin equation has led to  $B = 1.05$  for a prolate ellipsoid and  $B = 1.26$  for a Gaussian chain, depending on the degree of correlations between intramolecular distributions of a pair of molecules. Olaj et al.<sup>18</sup> have reported the variation of the molecular radius with the separation distance of two molecules.

Recent hydrodynamic studies of hard spherical particles have strongly supported the value  $B = 1.6$ , theoretically<sup>6,19</sup> and experimentally.<sup>20</sup> In the range  $M_w > 2 \times 10^6$ , the  $k_D^\Theta$  data in Figure 1 seem to meet this requirement (line M). In the lower  $M_w$  region, deviation from this relation may occur. At the present stage, we cannot judge whether the data really reject the  $M^{1/2}$  dependence over the entire  $M$  region as shown by the broken line in Figure 1 or show only a gradual change of the  $B$  value in eq 3 from one scheme to another, keeping the  $M^{1/2}$  dependence as required from the theories. Recently, a difference of about 15% in  $D_0$  between experimental<sup>1,21,22</sup> and theoretical<sup>17,23,24</sup> values at the  $\Theta$  temperature has definitely become clear. A novel theoretical development on the hydrodynamic interactions in dilute polymer solutions should be required.

**Acknowledgment.** We thank Prof. M. Kurata for his encouragement.

**Registry No.** Polystyrene (homopolymer), 9003-53-6.

## References and Notes

- Tsunashima, Y.; Nemoto, N.; Kurata, M. *Macromolecules* **1983**, *16*, 1184.
- Pyun, C. W.; Fixman, M. *J. Chem. Phys.* **1964**, *41*, 937.
- (a) Yamakawa, H. "Modern Theory of Polymer Solutions"; Harper and Row: New York, 1971. (b) Yamakawa, H. *J. Chem. Phys.* **1962**, *36*, 2995.
- Imai, S. *J. Chem. Phys.* **1970**, *52*, 4212.
- Mulderije, J. J. H. *Macromolecules* **1980**, *13*, 1526. The  $B$  values in the text, which are the special ones at the  $\Theta$  temperature, become 7.16 (Pyun-Fixman),<sup>2</sup> 7.01 (Mulderije),<sup>5</sup> and 6.55 (Batchelor)<sup>6</sup> at the hard-sphere (=good solvent) limit.
- Batchelor, G. K. *J. Fluid Mech.* **1972**, *52*, 245.
- Han, C. C. *Polymer* **1979**, *20*, 259.
- King, T. A.; Knox, A.; Lee, W. I.; McAdam, J. D. G. *Polymer* **1973**, *14*, 151.
- Jones, G.; Caroline, D. *Chem. Phys.* **1979**, *37*, 187.
- Frost, R. A.; Caroline, D. *Macromolecules* **1977**, *10*, 616.
- Homma, T.; Kawahara, K.; Fujita, H.; Ueda, M. *Makromol. Chem.* **1963**, *67*, 132.
- Wales, M.; Rehfeld, S. J. *J. Polym. Sci.* **1962**, *62*, 179.  $k_D^\Theta$  were roughly estimated from the graph (Figure 1) therein.
- Billick, I. H. *J. Phys. Chem.* **1962**, *66*, 1941.
- Kotaka, T.; Donkai, N. *J. Polym. Sci., Part A-2* **1968**, *6*, 1457.
- Mulderije, J. J. H. *Macromolecules* **1980**, *13*, 1207.
- McIntyre, D.; Wims, A.; Williams, L. C.; Manderkern, L. *J. Phys. Chem.* **1962**, *66*, 1932. The datum (only one point for  $M_w = 4.03 \times 10^6$ ) is not plotted in Figure 1 because of its too high value.
- Akcasu, A. Z. *Polymer* **1981**, *22*, 1169.
- Olaj, O. F.; Lantschbauer, W.; Pelinka, K. H. *Macromolecules* **1980**, *13*, 299.
- Felderhof, B. U. *J. Phys. A* **1978**, *11*, 929.
- Kops-Werkhoven, M. M.; Fijnaut, H. M. *J. Chem. Phys.* **1981**, *74*, 1618.
- Han, C. C.; Akcasu, A. Z. *Macromolecules* **1981**, *14*, 1080.
- Schmidt, M.; Burchard, W. *Macromolecules* **1981**, *14*, 201.
- Fixman, M. *Macromolecules* **1981**, *14*, 1710.
- Zimm, B. H. *Macromolecules* **1980**, *13*, 592.